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First-principles calculations of $GaAs_{1-x}P_x-Al_{0.3}Ga_{0.7}As(001)$ band offsets

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Abstract. The GaAs_{1-x}P_x-Al_{0.3}Ga_{0.7}As(001) band offsets are calculated for both the Ga-As and (Al_{0.3}Ga_{0.7})–(As_{1-x}P_x) interface bonds within first-principles local density functional theory. It is found that the band offsets are independent of the type of interfacial bond, while the valence band offset, VBO, decreases linearly with *x* as VBO = 0.143 - 0.153x eV and the inferred conduction band offset CBO (the band-gap difference minus the valence band offset) decreases linearly as CBO = 0.236 - 0.590x eV for $0.0 \le x \le 0.20$.

The GaAsP-GaAlAs strained-layer quantum well grown on a GaAs substrate is of technological significance, mainly because of its large optical nonlinearity. Zhang et al [1] have demonstrated that the excitonic transition peaks can achieve an energy blue-shift as large as 33 meV during photoluminescence measurements. Thus the GaAsP-GaAlAs strainedlayer quantum well is a good candidate for making optoelectronic devices such as optical switches and modulators. An accurate determination of the band offset at the GaAsP-GaAlAs strained-layer quantum well interfaces is very important for understanding its optical and transport properties. However, such a determination is complicated by the fact that, even for ideally abrupt interfaces, there are two possible different terminations: Ga-As and $(Al_{0.3}Ga_{0.7}) - (As_{1-x}P_x)$ interface bonds. To our knowledge, neither experimental data nor calculated values of the band offset for this system have been obtained so far. In this work, we present first-principles pseudopotential virtual-crystal calculations of the $GaAs_{1-x}P_x$ - $Al_{0.3}Ga_{0.7}As(001)$ band offsets, for both the Ga–As and $(Al_{0.3}Ga_{0.7})-(As_{1-x}P_x)$ interface bonds for x = 0.04, 0.08, 0.12, 0.16, 0.20. We find that the Ga-As and $(Al_{0.3}Ga_{0.7})$ - $(As_{1-x}P_x)$ interfaces have the same band offset, with both the valence band offset and conduction band offset decreasing linearly with the phosphorus composition x over the range $0.0 \leq x \leq 0.20$.

The valence band offset (VBO) can be written as the sum of two terms [2–6]: the bandstructure contribution ΔE_v , which is the difference between the energies of the valence band edges of the two bulk materials referred to the average electrostatic potential in the corresponding bulk material; and the electrostatic potential line-up ΔV , due to the chargedensity distribution at the interface. Each term is calculated within the framework of the density functional theory [7, 8] with the local density approximation (DFT–LDA), applied in the momentum-space formalism [9], using nonlocal norm-conserving pseudopotentials [10], and a plane-wave basis set. For the exchange–correlation potential we use the Ceperley– Alder [11] form of the LDA as parametrized by Perdew and Zunger [12]. The Brillouin zone

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(BZ) integrations are performed by sampling on a regular Monkhorst–Pack (MP) mesh [13] in reciprocal space, which is equivalent to using Chadi–Cohen special points [14]. Dandrea *et al* [15, 16] have summarized the manner in which band offsets are extracted from first-principles pseudopotential-based local density functional calculations (including spin–orbit effects), and the manner in which noncommon-atom heterojunctions allow for a possible dependence of the band offset on the interfacial atomic structure. We follow their scheme in our calculations. In addition, the virtual-crystal approximation (VCA) is used to describe the alloys of GaAsP and AlGaAs. It has been shown that the first-principles virtual-crystal calculations can yield a good description of the band offset in alloys [17, 18].

In order to calculate the GaAs_{1-x}P_x-Al_{0.3}Ga_{0.7}As(001) valence band offsets, we perform calculations on $(GaAs_{1-x}P_x)_5(Al_{0.3}Ga_{0.7}As)_4$ superlattices. Two such superlattices are studied: one with Ga-As interface bonds and the other with $(Al_{0.3}Ga_{0.7})-(As_{1-x}P_x)$ interface bonds. The atomic structure of the Ga-As-type superlattice is given by

$$\cdots (GaAs_{1-x}P_x)_5Ga\cdots As(Al_{0.3}Ga_{0.7}As)_4\cdots$$
(1)

while that of the $(Al_{0.3}Ga_{0.7})-(As_{1-x}P_x)$ type of superlattice is given by

$$\cdots (As_{1-x}P_xGa)_5(As_{1-x}P_x)\cdots (Al_{0.3}Ga_{0.7})(AsAl_{0.3}Ga_{0.7})_4\cdots .$$
(2)

Since the band offsets appropriate for strained $GaAs_{1-x}P_x$ on unstrained $Al_{0.3}Ga_{0.7}As$ (grown on a GaAs substrate) are of interest, the superlattice in-plane lattice constant a_{xy} is set equal to that of GaAs, i.e. $a_{xy} = a_{GaAs}$ (ignoring the difference between the lattice constant of GaAs and that of AlAs). As $GaAs_{1-x}P_x$ has a lattice constant smaller than that of $Al_{0.3}Ga_{0.7}As$, it will undergo a tetragonal distortion, as will the Ga–As and $(Al_{0.3}Ga_{0.7})-(As_{1-x}P_x)$ interface bonds.

The derivation of the electrostatic potential line-up, ΔV , at the interface requires that one first determine the positions of the atoms of minimum energy, i.e. the structure of the interface. This should be treated on the level of a quantum-mechanical energy minimization of the system with respect to the various parameters (strains and atomic displacements). Such a study has been performed for the prototypical case of a Si–Ge heterojunction by Van de Walle *et al* [6]. Their major conclusion is that the structure of the interface can, to a very good approximation, be determined using macroscopic elasticity theory. Furthermore, the small deviations of the minimum-energy structure from the ideal structure determined by the macroscopic elasticity theory give rise to only very small changes in the calculated band offsets. Thus, in our calculations, the positions of the atoms were determined by constructing ideal interfaces following the simple macroscopic elasticity rules. No further attempt was made to actually minimize the total energy.

Table 1. A summary of the GaAs_{1-x} P_x -Al_{0.3}Ga_{0.7}As(001) valence band offsets, conduction band offsets, and the corresponding ratios CBO/ ΔE_g for various phosphorus compositions.

P composition	VBO (eV)	CBO (eV)	$\mathrm{CBO}/\Delta E_\mathrm{g}$
0.04	0.137 ± 0.020	0.213 ± 0.020	0.61
0.08	0.131 ± 0.020	0.189 ± 0.020	0.58
0.12	0.125 ± 0.020	0.166 ± 0.020	0.57
0.16	0.120 ± 0.020	0.141 ± 0.020	0.54
0.20	0.112 ± 0.020	0.119 ± 0.020	0.52

Combining the band-structure term and the potential term, the valence band offset can be obtained. Convergence tests were performed with different supercell sizes, MP meshes, and kinetic energy cut-offs, and the relative numerical accuracy was estimated to be about 0.02 eV. The conduction band offset (CBO) can be derived from the relationship of $\Delta E_g = \text{VBO} + \text{CBO}$. However, the determination of the conduction band offset is not so straightforward in the present case, since the band gap is underestimated within the LDA, and the band gap of $\text{GaAs}_{1-x}P_x$ alloy is subject to changes due to strain. Thus we use the experimental relationship [19] below to describe the energy band gap of $\text{Al}_y\text{Ga}_{1-y}\text{As}$:

$$E_{\rm g} = 1.424 + 1.247y \qquad (0.0 \leqslant y \leqslant 0.45). \tag{3}$$

On the other hand, the band gap of $GaAs_{1-x}P_x$ without strain can be expressed as [20]

$$E_{\rm g} = 1.424 + 1.150x + 0.176x^2. \tag{4}$$

However, the strain in the GaAs_{1-x}P_x layer induces shifts in the valence and conduction bands which affect the band gap. To derive the conduction band offset in the present pseudomorphic strained-layer system, a knowledge of the strain-induced changes in the band gap is necessary. We assume that the DFT–LDA is able to give the correct strain-induced band-gap changes, on the basis of the fact that the DFT–LDA works well in calculations of deformation potentials, although it underestimates the band gap. The strain-induced bandgap changes can straightforwardly be extracted form the bulk band-structure calculations. Thus, the band gap of GaAs_{1-x}P_x alloy under strain can be obtained, and the conduction band offset can be inferred from the energy band-gap difference by subtracting the valence band offset. The calculated valence band offsets, the inferred conduction band offsets, and the corresponding ratios CBO/ ΔE_g are summarized in table 1, where ΔE_g is the band-gap difference.



Figure 1. The valence and conduction band offsets of $GaAs_{1-x}P_x-Al_{0.3}Ga_{0.7}As(001)$ heterostructures with various phosphorus compositions. The linear fitted lines are also plotted.

We find that the chemically abrupt $GaAs_{1-x}P_x-Al_{0.3}Ga_{0.7}As(001)$ interfaces have the same band offset, irrespective of the type of interfacial bond (Ga–As or (Al_{0.3}Ga_{0.7})–(As_{1-x}P_x)). This result is in agreement with that obtained by Dandrea and co-workers for a variety of heterostructures [15]. Figure 1 shows the calculated valence band offsets and the corresponding conduction band offsets as functions of the phosphorus composition in GaAs_{1-x}P_x. For the phosphorus compositions with *x* such that $0.0 \le x \le 0.20$, GaAs_{1-x}P_x–Al_{0.3}Ga_{0.7}As(001) heterostructures have a type I alignment, with the top of the valence band

of $GaAs_{1-x}P_x$ lying higher than that of $Al_{0.3}Ga_{0.7}As$, and the conduction band bottom lower than that of $Al_{0.3}Ga_{0.7}As$, Furthermore, the valence band offset, VBO, decreases linearly with x as VBO = 0.143 - 0.153x eV, while the conduction band offset decreases linearly with x as CBO = 0.2364 - 0.590x eV.

In conclusion, the first-principles pseudopotential method combined with the virtualcrystal approximation has been used to study the $GaAs_{1-x}P_x-Al_{0.3}Ga_{0.7}As(001)$ band offsets. The band offset is found to be independent of the type of interfacial bond (Ga–As or $(Al_{0.3}Ga_{0.7})-(As_{1-x}P_x))$, while both the valence band offset and the conduction band offset decrease linearly with the phosphorus composition over the range $0.0 \le x \le 0.20$.

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References

- [1] Zhang X, Karaki K, Yaguchi H, Onabe K, Ito R and Shiraki Y 1995 Appl. Phys. Lett. 66 186
- [2] Van de Walle C G and Martin R M 1985 J. Vac. Sci. Technol. B 3 1256
 Van de Walle C G and Martin R M 1986 J. Vac. Sci. Technol. B 4 1055
 Van de Walle C G and Martin R M 1986 Phys. Rev. B 34 5621
 Van de Walle C G and Martin R M 1987 Phys. Rev. B 35 8154
- [3] Baldereschi A, Baroni S and Resta R 1988 Phys. Rev. Lett. 61 734
- [4] Baroni S, Resta R, Baldereschi A and Peressi M 1989 Spectroscopy of Semiconductor Microstructures (NATO Advanced Study Institute, Series B: Physics, vol 106) ed G Fasol, A Fasolino and P Lugli (New York: Plenum) pp 251, 271
- [5] Resta R, Colombo L and Baroni S 1990 Phys. Rev. B 41 12358 (erratum 43 14273)
- [6] Colombo L, Resta R and Baroni S 1991 Phys. Rev. B 44 5572
- [7] Hohenberg P and Kohn W 1964 Phys. Rev. 136 B864
- [8] Kohn W and Sham L J 1965 Phys. Rev. 140 A1133
- [9] Ihm J, Zunger A and Cohen M L 1979 J. Phys. C: Solid State Phys. 12 4409
- [10] Hamann D R, Schlüter M and Chiang C 1979 Phys. Rev. Lett. 43 1494
- [11] Ceperley D M and Alder B 1980 Phys. Rev. Lett. 45 566
- [12] Perdew J P and Zunger A 1981 Phys. Rev. B 23 5048
- [13] Monkhorst H J and Pack J D 1976 Phys. Rev. B 13 5188
- [14] Chadi D J and Cohen M L 1973 Phys. Rev. B 8 5747
- [15] Dandrea R G, Duke C B and Zunger A 1992 J. Vac. Sci. Technol. B 10 1744
- [16] Dandrea R G and Duke C B 1993 Appl. Phys. Lett. 63 1795
- [17] Nelson J S, Wright A F and Fong C Y 1991 Phys. Rev. B 43 4908
- [18] Chua S J, Zhang X H, Xu S J and Gu X 1997 J. Phys.: Condens. Matter 9 L279
- [19] Adachi S 1985 J. Appl. Phys. 58 R1
- [20] Thompson A G, Cardona M, Shaklee K L and Woolley K L 1966 Phys. Rev. 146 601